

High thermal expansion KAlSiO_4 ceramic

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Orthorhombic kalsilite (KAlSiO_4) was prepared by solid-state reaction from K_2CO_3 , Al_2O_3 , and SiO_2 . The axial thermal expansion coefficients of the orthorhombic kalsilite were $1.6 \times 10^{-5} \text{ }^\circ\text{C}^{-1}$ for the *a*-axis, $1.6 \times 10^{-5} \text{ }^\circ\text{C}^{-1}$ for the *b*-axis, $2.8 \times 10^{-5} \text{ }^\circ\text{C}^{-1}$ for the *c*-axis, and $2.0 \times 10^{-5} \text{ }^\circ\text{C}^{-1}$ for the average from room temperature to $1000 \text{ }^\circ\text{C}$. A high thermal expansion ceramic consisting of the orthorhombic kalsilite was prepared by sintering. The densification was promoted by adding Li_2CO_3 . The KAlSiO_4 ceramic sintered at $1200 \text{ }^\circ\text{C}$ for 2 h with 5 wt % Li_2CO_3 had a bending strength of 65 MPa and linear thermal expansion coefficient of $2.2 \times 10^{-5} \text{ }^\circ\text{C}^{-1}$ from room temperature to $600 \text{ }^\circ\text{C}$.

1. Introduction

Although thermal expansions of ceramics are generally lower than those of metals, some framework silicates such as quartz (SiO_2), nepheline ($\text{NaAlSi}_3\text{O}_8$), sodalite ($\text{Na}_8\text{Cl}_2\text{Al}_6\text{Si}_6\text{O}_{24}$) and leucite (KAlSi_2O_6), show very high thermal expansions ($1.6\text{--}2.8 \times 10^{-5} \text{ }^\circ\text{C}^{-1}$) up to the α to β phase transformation [1–3]. Their high thermal expansions are interpreted as being due to the effect of the rotation of the framework (Si, Al) $_4$ tetrahedra from a partially collapsed state towards the fully expanded state and/or the untwisting of the collapsed frameworks [1–3]. For bonding to metals, we have studied how to prepare high thermal expansion ceramics from framework silicates. Previously, we reported the preparation of a high thermal expansion polycrystalline nepheline ceramic ($1.6 \times 10^{-5} \text{ }^\circ\text{C}^{-1}$) for bonding to iron or stainless steel and leucite ceramic ($2.4 \times 10^{-5} \text{ }^\circ\text{C}^{-1}$) for bonding to aluminium [4–6]. In this study, a high thermal expansion ceramic has been prepared from kalsilite (KAlSiO_4) for bonding on copper ($1.8 \times 10^{-5} \text{ }^\circ\text{C}^{-1}$) or silver ($2.2 \times 10^{-5} \text{ }^\circ\text{C}^{-1}$).

The structure of kalsilite is based upon a tridymite-type framework similar to nepheline [7]. The hexagonal kalsilite inverts at approximately $850 \text{ }^\circ\text{C}$ to an orthorhombic phase, which has a melting point of $1686 \pm 5 \text{ }^\circ\text{C}$ [7]. The inversion orthorhombic \rightleftharpoons hexagonal is sluggish, and the high-temperature orthorhombic form can be preserved at normal temperatures by quenching [7]. Thus, the thermal expansion and sintering of the orthorhombic kalsilite have been investigated in this work.

2. Experimental procedure

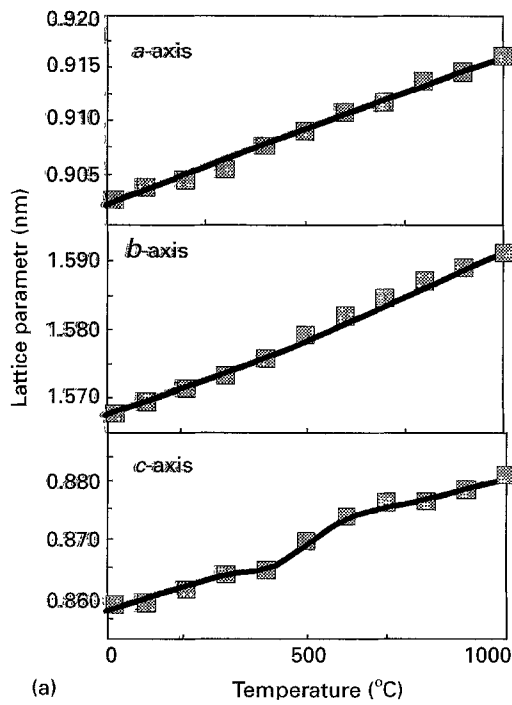
Orthorhombic kalsilite was prepared by solid-state reactions from reagent-grade potassium carbonate,

activated alumina, and silicon oxide. These raw materials were weighed to a 1:1:2 ratio of $\text{K}_2\text{O}:\text{Al}_2\text{O}_3:\text{SiO}_2$, dry-mixed in alumina mortar, calcined at $900 \text{ }^\circ\text{C}$ for 2 h, fired at $1300 \text{ }^\circ\text{C}$ for 48 h. The orthorhombic kalsilite powder was ground to an average particle size of approximately $1 \mu\text{m}$, and in some runs 1–10 wt % of an alkali carbonate (Li_2CO_3 , Na_2CO_3 or K_2CO_3) was added as a sintering aid. Test specimens were pressed at 100 MPa to form tablets of 16 mm diameter and $\sim 3 \text{ mm}$ height and rectangular bars of $60 \text{ mm} \times 5 \text{ mm}$. The specimens were then sintered at temperatures ranging from $1100\text{--}1400 \text{ }^\circ\text{C}$ for 2 h at a heating and cooling rate of $100 \text{ }^\circ\text{C h}^{-1}$.

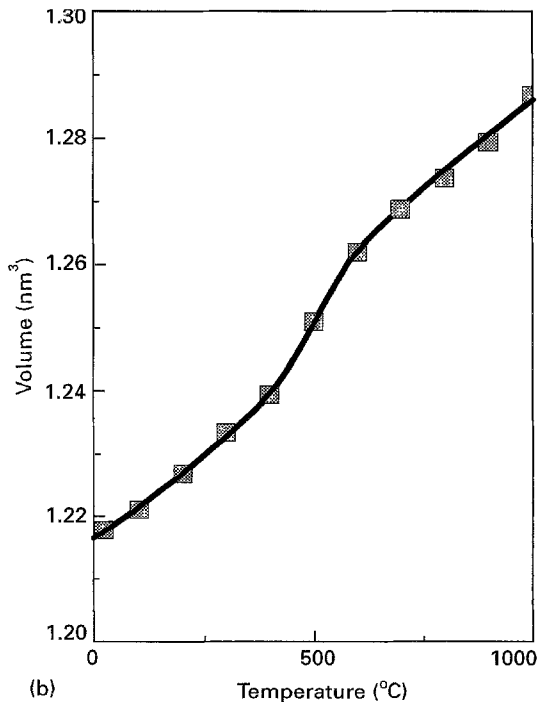
Bulk density was calculated from the weight and volume of the sintered body. Fracture strength was measured by the three-point bending method with a test span of 20 mm at a loading rate of 0.5 mm min^{-1} . The fracture surface was observed with a scanning electron microscope (JSM-T20, Nihon Denshi Co., Tokyo, Japan). The thermal expansion of ceramics was measured from room temperature to $600 \text{ }^\circ\text{C}$ at a heating rate of $10 \text{ }^\circ\text{C min}^{-1}$ using a dilatometer (TMA DT-30, Shimadzu Corp., Kyoto, Japan). The sintered samples were characterized by an X-ray powder diffraction, and the lattice parameters up to $1000 \text{ }^\circ\text{C}$ were determined using an X-ray diffractometer attached to a high-temperature apparatus (RINT1100, Rigaku Co., Tokyo, Japan). Platinum powder was employed as an internal standard. The diffraction patterns were measured in a 2θ range of $20^\circ\text{--}55^\circ$ using $\text{CuK}\alpha$ radiation with a scanning rate of $2^\circ 2\theta \text{ min}^{-1}$. The lattice parameters at each elevated temperature were calculated by using a least-squares computer program (UNICS RSLC-3) from eight major peaks, (002), (221), (041), (230), (202), (132), (311), and (241).

3. Results and discussion

Fig. 1 shows the thermal expansion of the crystal lattice parameters for the orthorhombic kalsilite. The *a*- and *b*-axes expanded almost linearly with increasing temperature up to 1000 °C. The *c*-axis had a thermal expansion which abruptly changed at 400–600 °C, conceivably resulting from the α to β phase transformation. The linear thermal expansion coefficients from room temperature to 1000 °C were $1.6 \times 10^{-5} \text{ } ^\circ\text{C}^{-1}$ for the *a*-axis, $1.6 \times 10^{-5} \text{ } ^\circ\text{C}^{-1}$ for the *b*-axis, and $2.8 \times 10^{-5} \text{ } ^\circ\text{C}^{-1}$ for the *c*-axis. The calculated average thermal expansion coefficient of the unit cell was $2.0 \times 10^{-5} \text{ } ^\circ\text{C}^{-1}$, which was intermediate between nepheline and leucite [4, 6].



(a) Temperature (°C)



(b) Temperature (°C)

Figure 1(a, b) Lattice parameters versus temperature for orthorhombic kalsilite.

When compacts of pure orthorhombic kalsilite powder were sintered at 1200–1400 °C, they did not densify. In order to prepare a dense ceramic, an alkali carbonate was added as a sintering aid. When

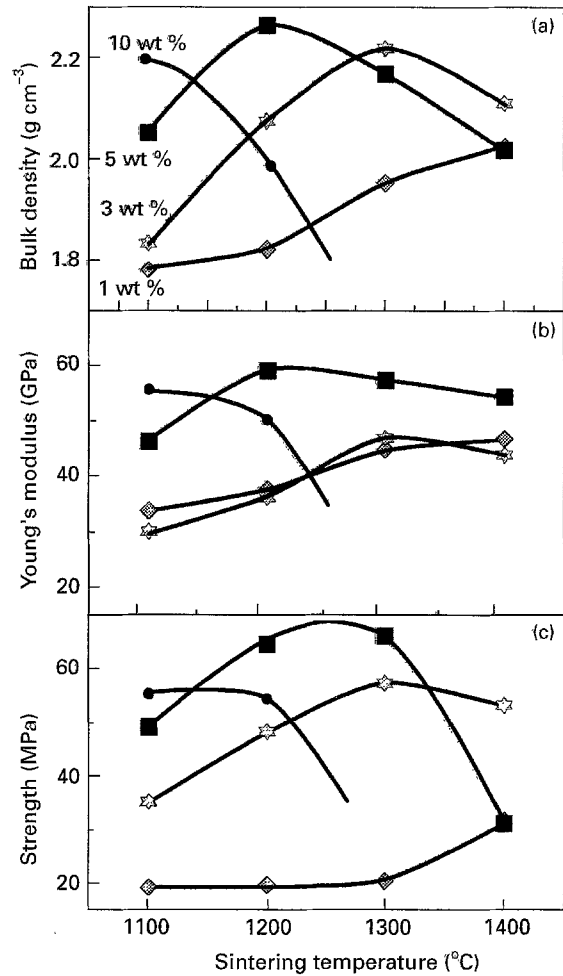


Figure 2 Variations of (a) bulk density, (b) Young's modulus and (c) bending strength with the amount of Li₂CO₃ and sintering temperature.

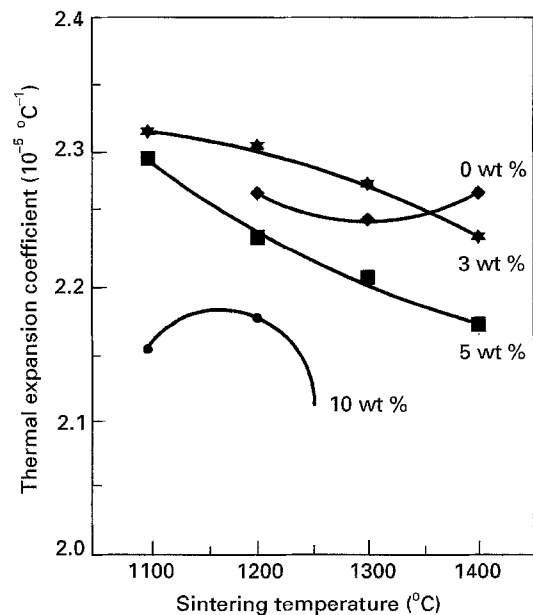


Figure 3 Variations of thermal expansion coefficient with the amount of Li₂CO₃ and sintering temperature.

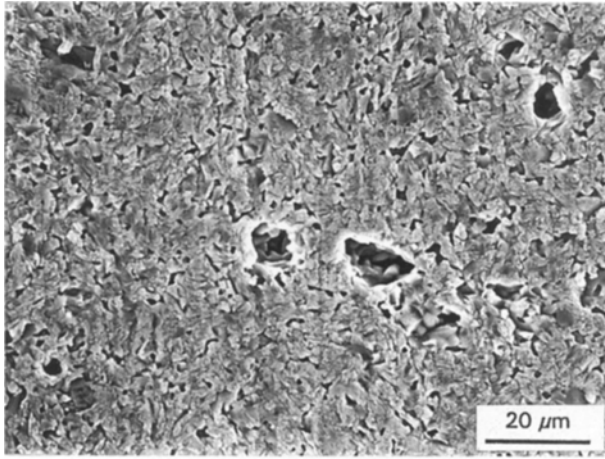


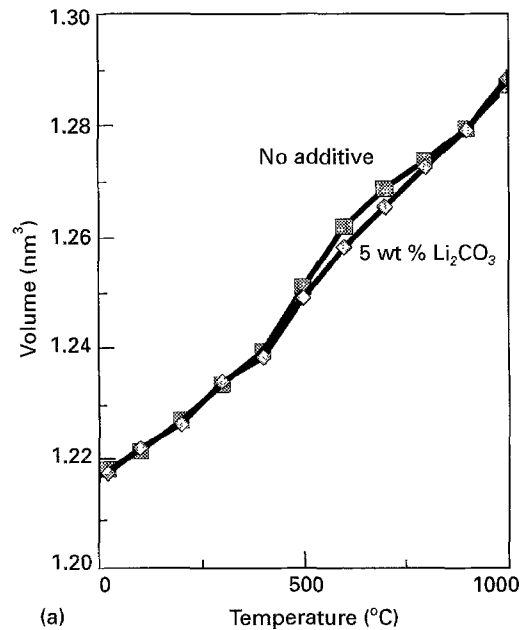
Figure 4 Scanning electron micrograph of a fracture surface of the orthorhombic kalsilite ceramic sintered at 1200 °C with 5 wt % Li_2CO_3 .

Na_2CO_3 or K_2CO_3 was added to the orthorhombic kalsilite powder, the specimens did not densify. On the other hand, when Li_2CO_3 was added, the specimens densified at lower temperatures in a manner similar to that of the nepheline ceramic [5]. A relative density of about 90% was achieved by sintering at 1200 °C for 2 h with 5 wt % Li_2CO_3 . At higher sintering temperatures, the specimens partly melted and then the relative density showed a downward trend.

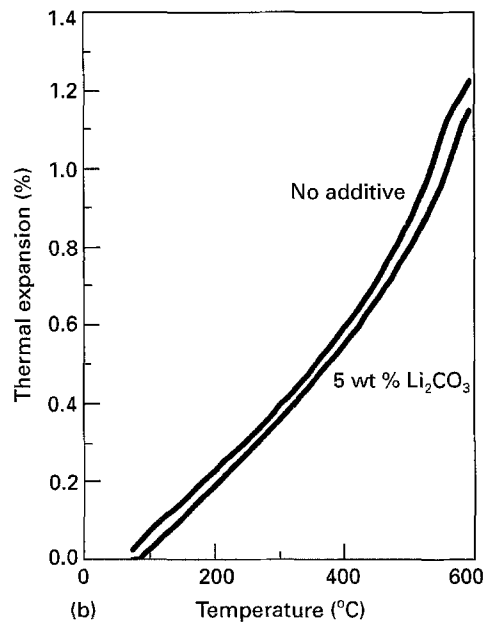
Fig. 2 shows the variations of bulk density, Young's modulus, and bending strength with the amount of Li_2CO_3 and sintering temperature. They increased with increasing amount of Li_2CO_3 and sintering temperature. However, they decreased with a larger amount of Li_2CO_3 and/or at a higher sintering temperature. The specimen completely melted on sintering at 1300 °C with 10 wt % of Li_2CO_3 . Fig. 3 shows the thermal expansion coefficients from room temperature to 1000 °C of the orthorhombic kalsilite ceramics. The thermal expansion coefficient decreased with increasing amount of Li_2CO_3 and sintering temperature. The specimen melted by sintering at 1300 °C with 10 wt % Li_2CO_3 had a lower thermal expansion coefficient of about $1.2 \times 10^{-5} \text{ }^\circ\text{C}^{-1}$. Consequently, it was considered that the lowering of density, Young's modulus, strength, and thermal expansion coefficient was due to the formation of a glass phase. Fig. 4 demonstrates that grains were surrounded by a glassy substance with some large pores.

In conclusion, the addition of Li_2CO_3 accelerated the formation of a liquid phase, resulting in faster densification at lower temperatures. The orthorhombic kalsilite ceramic obtained by sintering at 1200 °C for 2 h with 5 wt % Li_2CO_3 revealed a bending strength of 65 MPa, and a Young's modulus of 60 GPa; its high thermal expansion coefficient, $2.2 \times 10^{-5} \text{ }^\circ\text{C}^{-1}$, was almost equal to the average of the unit cell.

In addition, Fig. 5 shows the variation of the thermal expansion curves of the orthorhombic kalsilite crystal and ceramic with the addition of Li_2CO_3 . When Li_2CO_3 was added to the orthorhombic kalsilite, the thermal expansion of both the crystal and ceramic altered very little. However, the abrupt expansion at 400–600 °C was somewhat smoothed. Consequently,



(a)



(b)

Figure 5 Variation of the thermal expansion curves of (a) orthorhombic kalsilite crystal and (b) ceramic with the addition of Li_2CO_3 .

it was considered that some Li^+ might be incorporated into the orthorhombic kalsilite crystal lattice.

References

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